

## CHAPTER IV

### IMPORTANCE OF THE SOLID STATE OF 17 $\beta$ -ESTRADIOL ON ITS RELEASE CHARACTERISTIC FROM IMPLANT USING EUDRAGIT<sup>®</sup> RS AS A RELEASE CONTROLLING AGENT

#### 4.1 Introduction

Over the last two decades, matrix system containing a poorly water-soluble drug and using an inert polymer as the matrix excipient has been widely investigated. Many researchers have indicated that it is possible to obtain a zero-order release kinetic from this controlling system (Chandrasekaran and Paul, 1982; Ford et al., 1987; El-Arini and Leuenberger, 1995; Kim, 2000b; Costa and Lobo, 2001; Siepmann and Peppas, 2001). Studies conducted by Ford et al. (Ford, Rubinstein, and Hogan, 1985a; 1985b; 1985c; Ford et al., 1987) reported that dissolution profiles of two insoluble drugs, indomethacin and diazepam, from hydroxypropylmethylcellulose (HPMC) matrices exhibited near zero-order release kinetics whereas dissolution profiles of soluble drugs, including promethazine hydrochloride, aminophylline, tetracycline hydrochloride and propranolol hydrochloride, from HPMC matrices exhibited square root of time kinetics. These studies corresponded to the one conducted by El-Arini and Leuenberger (1995) which indicated the zero-order release kinetic of caffeine representing a poorly water-soluble drug from polyvinylpyrrolidone (PVP) matrix and the square root of time kinetic of potassium chloride representing a water-soluble drug from the same polymeric matrix. Chandrasekaran and Paul (1982) indicated that the mass released had a square-root dependency with time in which dispersed matrix was totally controlled by diffusion process. On the contrary, the mass released varied directly with time or the release rate of a dispersed drug would become time independent in which drug dissolution offered the limiting resistance to mass transport. From previous chapter, the release kinetics of 17 $\beta$ -estradiol (E<sub>2</sub>) and norethindrone (NET) from implants using Eudragit<sup>®</sup> RS (ERS) as a release controlling agent exhibited zero-order release kinetics. The obtained release kinetics did not result from the Geomatrix<sup>®</sup> design. Kim (2000b) indicated that geometry was not an important factor for a drug dissolution controlled release system. Furthermore, the porosity and the tortuosity did not play the leading role in controlling E<sub>2</sub> and NET released. In case of poorly water-soluble drug, dissolved drug and non-dissolved drug coexist within the polymer matrix. Siepmann and Peppas (2001) indicated that non-dissolved drug was not available for diffusion. Non-dissolved drug remaining within the polymer matrix resulted in drug concentration gradient unchanged and the absolute amount of drug released within a certain time period remained constant. This non-dissolved drug effect overcompensated the previously described porosity and geometry effect.

For polymer matrix containing poorly water-soluble drug, drug solubility under a given condition plays the leading role in controlling drug release. In the same way, the inherent solubility of E<sub>2</sub> under a given condition is a major factor affecting

release profile and release kinetics of  $E_2$  implant using ERS as a release controlling agent. If the solubility of  $E_2$  in the polymer matrix under a given condition changes, release profile and release kinetics of  $E_2$  should be altered. In polymer matrix containing drug at concentration lower than its solubility in that polymer, drug exists in a dissolved state. On the contrary, drug exists in a crystalline state when its concentration in the polymer matrix is much higher than its solubility in that polymer. Patterson et al. (2006) indicated that an amorphous drug existed where drug was dispersed in the polymer on a molecular level. The solubility of a compound in the amorphous form is higher than that in the more stable crystalline form because the Gibbs free energy is higher (Martin, 1993). The dissolution rate of an amorphous compound is improved relative to a crystalline form (Patterson et al., 2006). Thus, release profile and release kinetics of amorphous  $E_2$  should be different from those of a crystalline  $E_2$ .

The aims of this study were

- (i) to determine the influence of solid state of  $E_2$  on the  $E_2$  release characteristic.
- (ii) and to investigate the effect of weight percent of  $E_2$  in ERS solid dispersions on the solid state of  $E_2$ .

## 4.2 Materials and Methods

### 4.2.1 Materials

17 $\beta$ -estradiol (E<sub>2</sub>) and Benzalkonium chloride (BAC) were purchased from Fluka Chemica, Germany. Eudragit<sup>®</sup> RS PO (Röhm Pharma GmbH, Germany) was donated by JJ Degussa, Thailand. Absolute ethanol and dichloromethane were of a reagent grade purchased from Merck, Germany. Acetonitril was of a HPLC grade purchased from Fisher Scientific (UK). Sodium hydroxide and potassium dihydrogen phosphate were obtained from Mallinckrodt (Mexico) and Asia Pacific Specialty Chemicals Limited (Australia), respectively.

### 4.2.2 Preparation and Determination of E<sub>2</sub> Content in Solid Dispersions

Solid dispersions containing 10, 20 and 30 % w/w E<sub>2</sub> in ERS were prepared and E<sub>2</sub> content in each sample was determined as described in previous study. Solid dispersions containing 1 and 2 % w/w E<sub>2</sub> in ERS were prepared with the same procedure as those containing E<sub>2</sub> at concentration range of 10-30 % w/w, except when a mixture of absolute ethanol: dichloromethane (1:1; v:v) was used in the process of dissolving E<sub>2</sub> and ERS at mass ratios of 1/99 and 2/98.

### 4.2.3 Preparation of Implants

Implants containing 1 or 2 % w/w E<sub>2</sub> in ERS solid dispersions were produced by the same method as that of implants containing 10, 20 and 30 % w/w E<sub>2</sub> in ERS solid dispersions as described in previous chapter.

### 4.2.4 Drug Release Study

Release studies of implants containing 1 and 2 % w/w E<sub>2</sub> in ERS solid dispersion were conducted, in triplicate, in the same way as that of implants containing 10, 20 and 30 % w/w E<sub>2</sub> in ERS solid dispersion as described in previous study. Release medium of implants containing 1 and 2 % w/w E<sub>2</sub> in ERS solid dispersion was taken out and replaced by fresh release medium at 1 hr, 1, 2, 3, 5 day, then the concentration of E<sub>2</sub> was assayed by validated HPLC with the same condition as previously described. After in vitro release study, each implant was dissolved in a mixture of absolute ethanol and dichloromethane (1:1; v:v) and the solution was assayed by UV-spectroscopy (Jasco V-530, Japan) at 280 nm for E<sub>2</sub> residual content. Total E<sub>2</sub> content of each implant was calculated by addition of total E<sub>2</sub> released and residual E<sub>2</sub> content in implant after in vitro release study.

### 4.2.5 Determination of Drug Release Kinetics

Approximately 60% of E<sub>2</sub> released from each implants was fitted with three different release models: the zero-order, the first-order, and the Higuchi model, by linear regression analysis. The coefficient of determination (R<sup>2</sup>) obtained from each fit was used as a criterion to choose the best model for drug release phenomena.

## 4.2.6 Characterization of E<sub>2</sub> Solid State in Solid Dispersions

### 4.2.6.1 Polarized Light Microscopy (PLM)

An Olympus BX51 polarized light microscope equipped with a Nikon E990 (Japan) digital camera and Image ProPlus software V4.0 (Media Cybernetics) was used. ERS, solid dispersions containing E<sub>2</sub> at 1, 2, 10, 20, 30 % w/w or E<sub>2</sub> crystal powder was brushed onto a glass slide and covered with a cover-slip. E<sub>2</sub> solid state in these samples was observed under polarized light plus purple light generator. The objective and the ocular magnification was 10x, so that the total magnification was equal to 100.

### 4.2.6.2 X-ray Powder Diffractometry (XRPD)

A Siemens D5000 diffractometer (Stuttgart, Germany) was used. E<sub>2</sub>, solid dispersions containing E<sub>2</sub> at concentration range of 1-75 % w/w, and ERS were scanned from 5° to 90° 2θ (sampling interval of 0.02°) using Ni-filtered Cu Kα radiation. Operating voltage and current were 40 kV and 30 mA, respectively.

### 4.2.6.3 Differential Scanning Calorimetry (DSC)

A DSC TA Q100 with a refrigerated cooling system (TA Instruments, New Castle, DE) and nitrogen as purge gas was used. The calorimeter was calibrated using indium and sapphire for temperature and heat capacity, respectively. E<sub>2</sub> of 3.17 mg was added to standard aluminum pan with cover and scanned using the following heating program: heating up to 182 °C at 10 °C/min; cooling down to 0 °C at 20 °C/min; heating up to 250 °C at 10 °C/min.

### 4.2.6.4 Thermogravimetric Analysis (TGA)

A TGA/SDTA 851e with a refrigerated cooling system (Mettler Toledo, Switzerland) and nitrogen as purge gas was used. E<sub>2</sub> was added in an open pan and scanned from 0-750 °C at heating rate of 5 °C/min. Weight loss was determined gravimetrically.

### 4.2.6.5 Modulated Temperature Differential Scanning Calorimetry (MTDSC)

A MTDSC TA Q100 with a refrigerated cooling system (TA Instruments, New Castle, DE) and nitrogen as purge gas was used. Each sample of E<sub>2</sub>, solid dispersion containing E<sub>2</sub> at concentration range of 1-90 % w/w, or ERS was added to standard aluminum pans with covers and scanned using two heating programs. Heating program I: heating from 25 to 120 °C at 10 °C/min; cooling down to 25 °C at 20 °C/min, an isothermal period for 5 min at 25 °C, and finally heating to 250 °C at 5 °C/min. Heating program II: heating from 25 to 182 °C at 10 °C/min, cooling down to 25 °C at 20 °C/min, an isothermal period for 5 min at 25 °C, and finally heating to 250°C at 5 °C/min. A modulation amplitude of ±1°C and a period of 60 s were used.

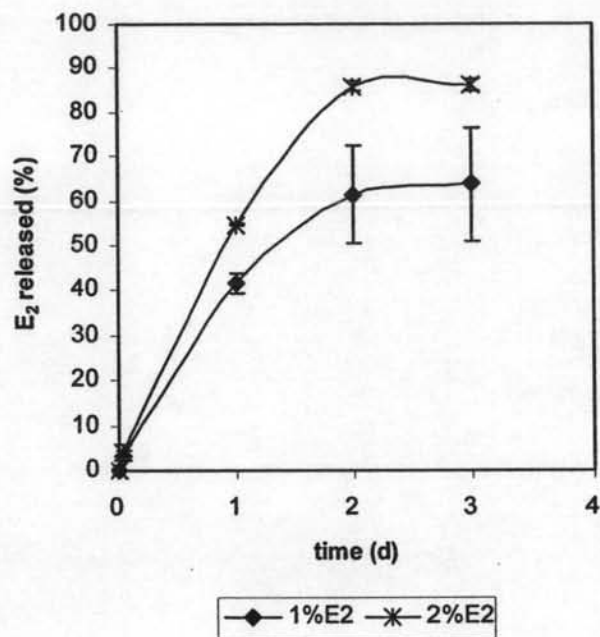
## 4.3 Results and Discussion

### 4.3.1 Effect of Weight Percent of E<sub>2</sub> in ERS Solid Dispersion on E<sub>2</sub> Release Profile

E<sub>2</sub> release profiles of implants containing 1 and 2 % w/w E<sub>2</sub> in ERS solid dispersion are shown in Figure 4.1 (a). E<sub>2</sub> was rapidly released from implants containing 1 and 2 % w/w E<sub>2</sub> in ERS solid dispersion. Approximately 60 % of E<sub>2</sub> were released from those implants within 3 days. E<sub>2</sub> release profiles of implants containing 10, 20 and 30 % w/w E<sub>2</sub> in ERS solid dispersion are shown in Figure 3.3. Percent cumulative releases of E<sub>2</sub> from these implants increased linearly with time. Approximately 60 % of E<sub>2</sub> were released from these implants within 5 days. This indicated that higher weight percent of ERS did not extend E<sub>2</sub> released for longer period. Figure 4.1 (b) illustrates E<sub>2</sub> daily release rates obtained from implants containing 1 and 2 % w/w E<sub>2</sub> in ERS solid dispersion. E<sub>2</sub> daily release rates from those implants drastically decreased with time whereas E<sub>2</sub> daily release rates obtained from implants containing 10, 20 and 30 % w/w E<sub>2</sub> in ERS solid dispersion were rather constant during 1-5 days as shown in Figure 3.4.

When approximately 60 % of experimental E<sub>2</sub> released from implants containing solid dispersions at different E<sub>2</sub> to ERS ratios were fitted with the zero-order, the first-order or the Higuchi release models by linear regression analysis, the coefficient of determination (R<sup>2</sup>) obtained from each fit is shown in Table 4.1. It is apparent that the zero-order model might not be a suitable model which could be used to describe the E<sub>2</sub> released from implants containing 1 and 2 % w/w E<sub>2</sub> in ERS solid dispersion compared with the first-order and the Higuchi models. The first-order model describes drug release in a way that is proportional to the amount of drug remaining in its interior, in such way that the amount of drug released by unit of time diminishes (Costa and Lobo, 2001). The Higuchi model describes drug release in a way that the fraction of drug released is proportional to the square root of time. Alternatively, the drug release rate is proportional to the reciprocal of the square root of time (Siepmann and Peppas, 2001). Thus, the release rate of controlled release system described by these two models would become time dependent. This corresponded to decrease E<sub>2</sub> daily release rates with time of implants containing 1 and 2 % w/w E<sub>2</sub> in ERS solid dispersion. Furthermore, the Higuchi model describes drug release based on diffusion mechanism. A proportionality between the cumulative amount of drug released and the square root of time is commonly regarded as an indicator for diffusion-controlled drug release (Siepmann and Peppas, 2001). Thus, E<sub>2</sub> release profiles obtained from implants containing 1 and 2 % w/w E<sub>2</sub> in ERS solid dispersion which could be described by the Higuchi model indicated that diffusion process was the dominating mechanism in these systems.

(a)



(b)

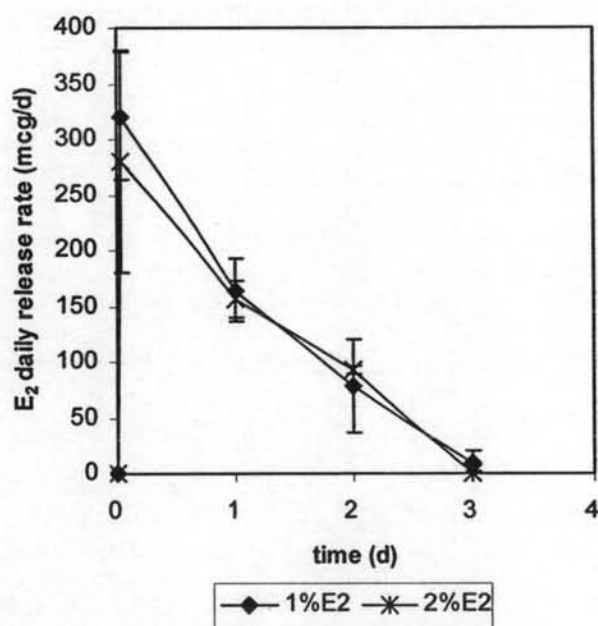


Figure 4.1  $E_2$  release profiles (a) and  $E_2$  daily release rates (b) from implants containing 1 and 2 % w/w  $E_2$  in ERS solid dispersion

**Table 4.1  $R^2$  obtained from fitting approximate 60 % of  $E_2$  released with the zero-order, the first-order, and the Higuchi models**

Weight percent $E_2$ in implant (n=3)	$R^2$		
	$Q_t = Q_0 + k_0t$	$Q_t = Q_0 \times \exp(-k_1t)$	$Q_t = k_H t^{1/2}$
1% $E_2$	0.9703	0.9977	0.9933
2% $E_2$	0.9821	0.9895	0.9897
10% $E_2$	0.9995	0.9835	0.9368
20% $E_2$	0.9980	0.9684	0.9320
30% $E_2$	0.9946	0.9567	0.8978

$Q_t$ , the amount of drug released at time  $t$ ;  $Q_0$ , the initial amount of drug released;  $k_0$ , the zero-order release rate constant;  $k_1$ , the first-order release rate constant;  $k_H$ , the Higuchi dissolution rate constant.

For implants containing 10, 20 and 30 % w/w  $E_2$  in ERS solid dispersion,  $R^2$  obtained from fitting 60 % of experimental  $E_2$  released with the zero-order model were the highest of all release models. The zero-order model describes the system releasing the same amount of drug by unit of time. This release model was in agreement with constant  $E_2$  daily release rates obtained from implants containing 10, 20 and 30 % w/w  $E_2$  in ERS solid dispersion during 1-5 days. This indicated that  $E_2$  daily release rates under this condition would become time independent. Kurnik and Potts (1997) used a combined dissolution- and diffusion-controlled release system for describing the effect of crystal particle size on the release of estradiol from polymer matrices in water. This system explains drug release following two processes. The drug first dissolves in the release medium in the matrix and then diffuses out of the matrix. When the dissolution rate is slower than the diffusion rate of the drug, the former rate markedly influences the drug release and the zero-order release can be obtained (Chandrasekaran and Paul, 1982; Kim, 2000b). This indicated that dissolution rate of  $E_2$  in implants containing 10, 20 and 30 % w/w  $E_2$  in ERS solid dispersion was the dominating mechanism. Furthermore, Chang and Himmelstein (1990) indicated that the zero-order release could not be obtained for system having a high dissolution rate constant of drug. These suggested that dissolution rate of  $E_2$  in implants containing 1 and 2 % w/w  $E_2$  in ERS solid dispersion was higher than that of  $E_2$  in implants containing 10, 20 and 30 % w/w  $E_2$  in ERS solid dispersion. The dissolution rate of  $E_2$  in implants containing 1 and 2 % w/w  $E_2$  in ERS solid dispersion did not offer as the limiting resistance to drug release, so that the zero-order release could not be achieved. On the contrary, the dissolution rate of  $E_2$  in implants containing 10, 20 and 30 % w/w  $E_2$  in ERS solid dispersion performed as the limiting resistance to drug release. Thus, 60 % of  $E_2$  release followed the zero-order kinetic.

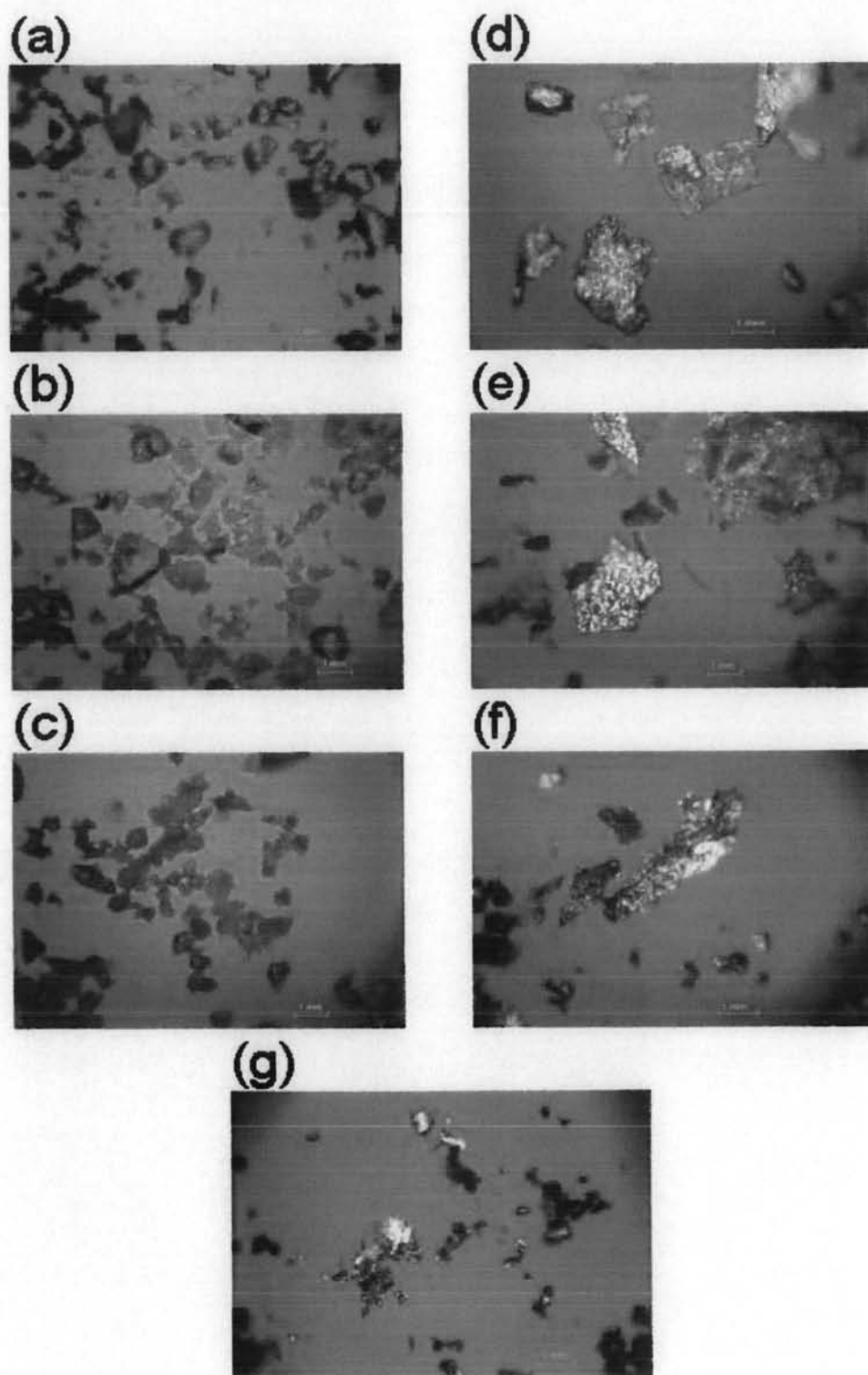
#### 4.3.2 Effect of Weight Percent of $E_2$ in ERS Solid Dispersion on $E_2$ Solid State

##### 4.3.2.1 $E_2$ Solid State Characterized by PLM

PLM was used to examine the solid state of  $E_2$  in solid dispersions. An amorphous solid observed under polarized light microscope normally manifests the absence of birefringence because the structure of an amorphous material usually possesses short-range molecular arrangement but lacks long-range translational-orientation symmetry and has regions of different density (Yu, 2001; Sacchetti and Zhu, 2002). Thus, amorphous solid cannot turn plane polarized light and cannot reflect purple light to other wavelengths in range of the spectrum. In any way, the birefringence is always observed in a crystalline material because the crystalline state exhibited regular pattern of molecular arrangements. The molecules in a crystal arranged in a fixed pattern in three dimensions known as a lattice (Gioielli, 2003). Thus, crystalline solid can turn plane polarized light and can reflect purple light to other wavelengths in range of spectrum.

Photomicrographs of ERS, solid dispersion containing 1, 2, 10, 20, 30 % w/w  $E_2$  in ERS and  $E_2$  under polarized light microscope are shown in Figure 4.2. The absence of birefringence was observed in ERS, 1 and 2 % w/w  $E_2$  in ERS solid





**Figure 4.2 Photomicrographs obtained from polarized light microscope:**  
(a) ERS; solid dispersion containing weight percent of E<sub>2</sub> at (b) 1; (c) 2; (d) 10;  
(e) 20; (f) 30; (g) E<sub>2</sub>

dispersion under polarized light microscope as shown in Figure 4.2 (a)-(c). This indicated that  $E_2$  was in an amorphous state in 1 and 2 % w/w  $E_2$  in ERS solid dispersion because these two solid dispersions could not turn plane polarized light and could not reflect purple light to other wavelengths in range of the spectrum. In comparison with 10, 20, and 30 % w/w  $E_2$  in ERS solid dispersions, the birefringence was observed under polarized light microscope as shown in Figure 4.2 (d)-(f). This phenomenon was also observed in  $E_2$  crystal powder as shown in Figure 4.2 (g). These solid dispersions could turn plane polarized light and could reflect purple light to other visible wavelengths as same as  $E_2$  crystal did, so that the solid state of  $E_2$  in these solid dispersions was similar to  $E_2$  crystal, that is, a crystalline state.

#### 4.3.2.2 $E_2$ Solid State Characterized by XRPD

X-ray powder diffraction patterns of ERS,  $E_2$ , and  $E_2$  in ERS solid dispersions at concentration range of 1-75 % w/w are shown in Figure 4.3. X-ray powder diffraction patterns of  $E_2$  exhibited characteristic peaks at a diffraction angle of  $2\theta$ , at 13.14, 15.74, 18.26, 22.62, and 26.58° as shown in Figure 4.3 (a). These values and the diffraction pattern were comparable to those reported by Latsch et al. (2003) and Park et al. (2005) characterizing  $E_2$ -hemihydrate crystal. These characteristic peaks were also observed in X-ray powder diffraction patterns of  $E_2$  in ERS solid dispersions at concentration range of 10-75 % w/w as shown in Figure 4.3 (b)-(f) but the relative peak intensity and the distinct X-ray pattern decreased in relation to the decrease in weight percent of  $E_2$ . However, these characteristic peaks could not be observed in X-ray powder diffraction patterns of 1 and 2 % w/w  $E_2$  in ERS solid dispersion as shown in Figure 4.3 (g)-(h). X-ray powder diffraction patterns of these two solid dispersions did not exhibit a distinct X-ray pattern as same as that of ERS. This is a nature of an amorphous solid. This result indicated an amorphous state of  $E_2$  in 1 and 2 % w/w  $E_2$  in ERS solid dispersion characterized by XRPD. Due to an amorphous nature of ERS, higher weight percent of ERS corresponding to lower weight percent of  $E_2$  in solid dispersion resulted in less crystalline fraction. Thus, the relative peak intensity of those characteristic peaks decreased as fraction of  $E_2$  crystal decreased. However, those characteristic peaks were observed in solid dispersions containing  $E_2$  at 10, 20 and 30 % w/w.  $E_2$  existed in a crystalline state in these concentrations. Therefore, the  $E_2$  solid state in solid dispersion examined by XRPD analysis agreed with that examined by PLM.

#### 4.3.2.3 $E_2$ Solid State Characterized by Thermal Analysis

##### 4.3.2.3.1 DSC and TGA

DSC curve of  $E_2$  is shown in Figure 4.4. In the first heating run (1), three endothermic peaks were observed. The first two endothermic peaks corresponded to the weight loss around 3.20 % as shown in the thermogravimetric (TGA) curve of  $E_2$  (Figure 4.5). The third endothermic peak at 179.24 °C was the melting point of  $E_2$ . This corresponded with that of  $E_2$ -hemihydrate assigned to EA form according to the Variankaval, Jacob, and Dinh (2000) study. The weight loss around 3.20 % indicated that the stoichiometry of  $E_2$  should be  $C_{18}H_{24}O_2 \cdot \frac{1}{2} H_2O$ . The DSC curve during cool down exhibited  $T_g$  of  $E_2$  around 84.20 °C corresponding to  $T_g$

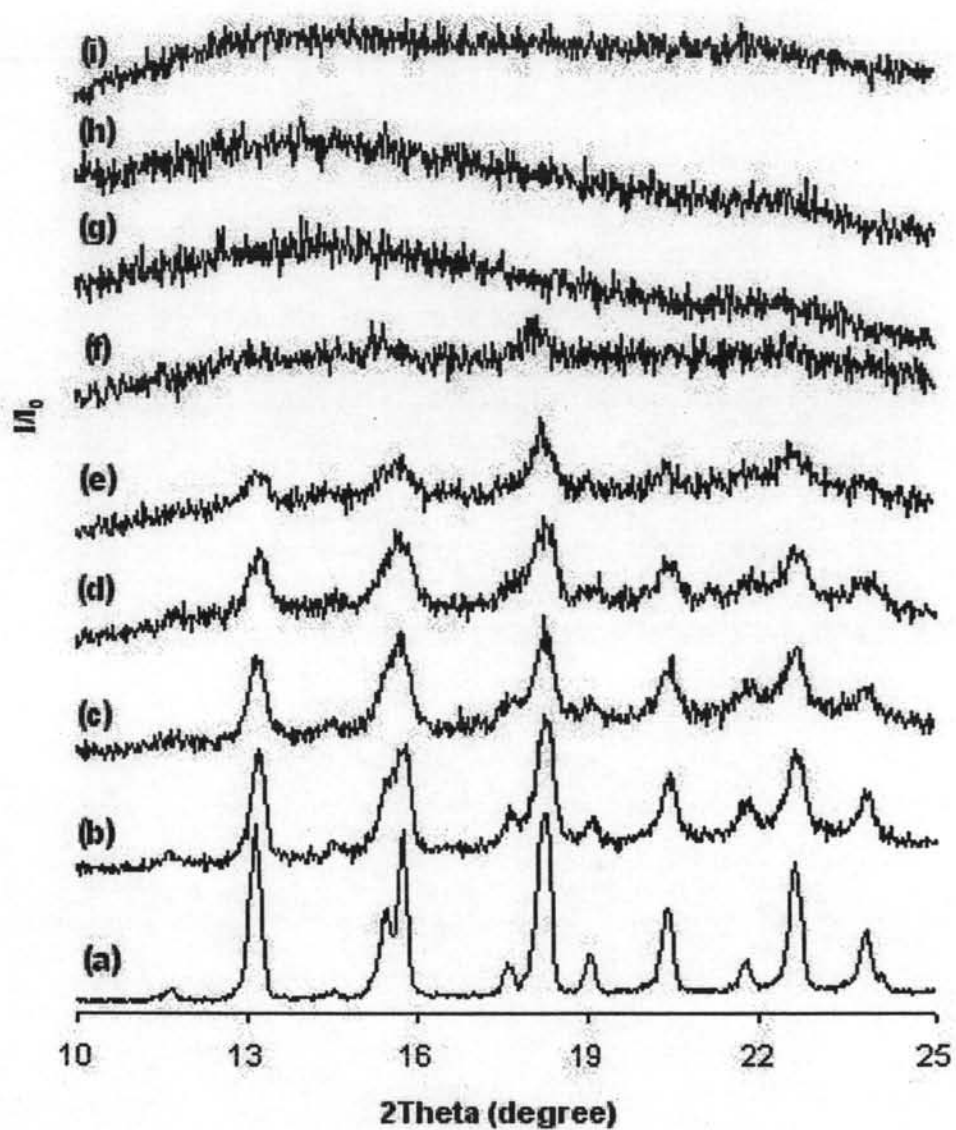
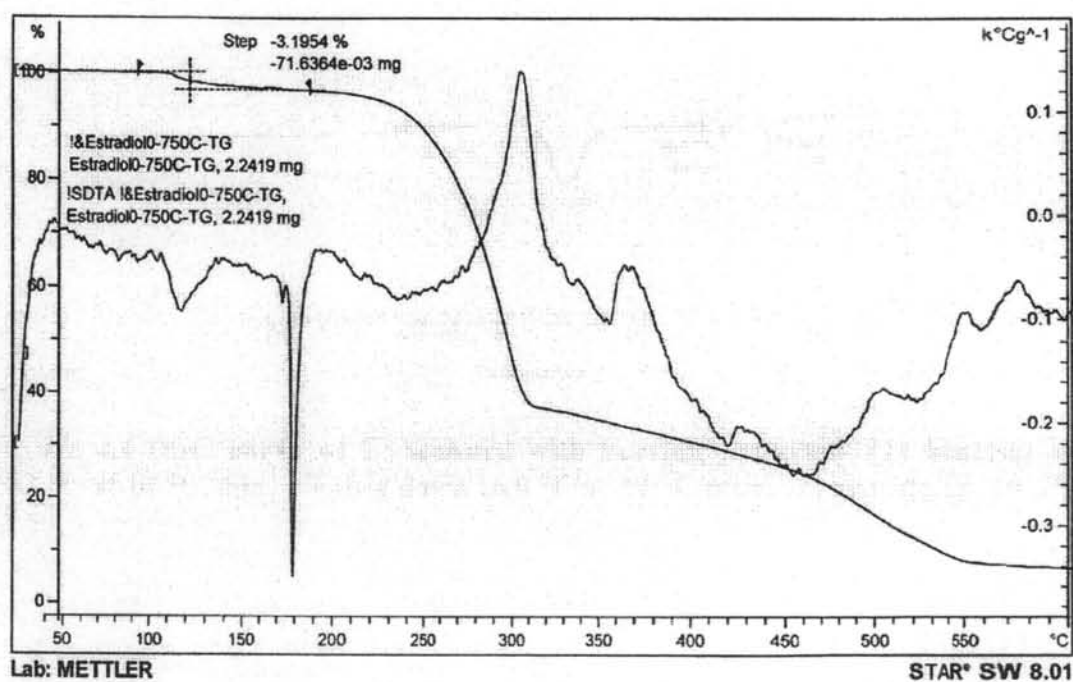
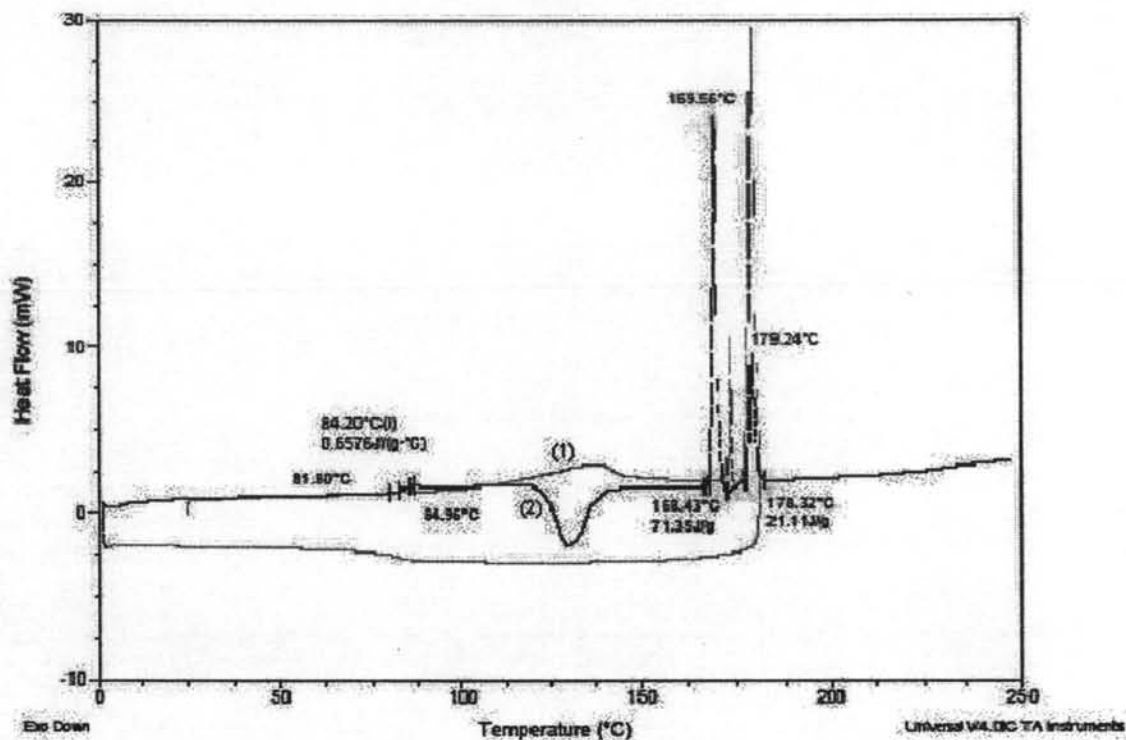


Figure 4.3 X-ray powder diffraction patterns of solid dispersions containing  $E_2$  at different weight percent: (a)  $E_2$ ; (b) 75 %  $E_2$ ; (c) 50 %  $E_2$ ; (d) 30 %  $E_2$ ; (e) 20 %  $E_2$ ; (f) 10 %  $E_2$ ; (g) 2 %  $E_2$ ; (h) 1 %  $E_2$ ; (i) ERS



of  $E_2$  observed in the second heating run (2). This indicated that  $E_2$  changed to an amorphous form after heated to 182 °C and then rapidly cooled down because of manifesting  $T_g$  from its amorphous nature. In the second heating run, the amorphous form of  $E_2$  in the glassy state was changed to the rubbery state when the temperature was higher than 80 °C. Exothermic peak in the second heating run was observed around 126 °C, followed by two endothermic peaks at 169.56°C and 179.24°C. This result agreed with that obtained by Variankaval et al. (2000) indicating that the first and the second endothermic peaks were assigned to ED form and EC form of  $E_2$ , respectively.

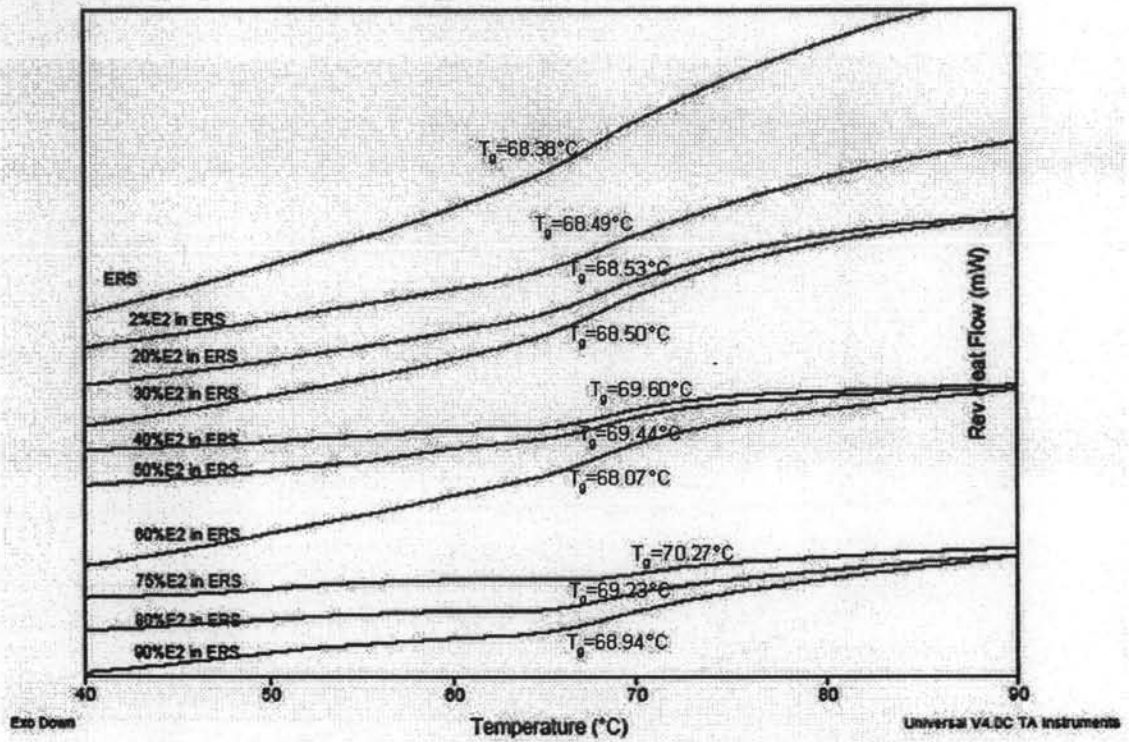
The result suggested that  $T_g$  of  $E_2$  could be observed when  $E_2$  existed in an amorphous form. In order to differentiate  $E_2$  solid state in solid dispersion in the following experiment, two indicators, i.e.  $T_g$  behavior of a blend and melting point of  $E_2$  in solid dispersion, were investigated. In solid dispersion containing amorphous  $E_2$ ,  $T_g$  should lie between those of  $E_2$  and ERS based on the behavior of an amorphous blend in multi-component system (Yu, 2001). In solid dispersion containing crystalline  $E_2$ ,  $T_g$  should belong to  $T_g$  of ERS only and the melting point of crystalline  $E_2$  should be observed.

#### 4.3.2.3.2 MTDSC

From MTDSC analysis with heating program I,  $T_g$  behavior of solid dispersion containing  $E_2$  at concentration range of 0-90 % w/w examined from reverse heat flows is shown in Figure 4.6 (a).  $T_g$  values of  $E_2$  in ERS solid dispersion at concentration range of 1-90 % w/w obtained from MTDSC using heating program I were not different from  $T_g$  value of ERS whereas  $T_g$  values of those obtained from MTDSC using heating program II increased toward  $T_g$  value of amorphous  $E_2$  when weight percent of  $E_2$  increased as shown in Figure 4.6 (b). The resulting  $T_g$  of blends obtained from MTDSC using heating program II were affected by  $T_g$  of amorphous  $E_2$  and ERS. This is the nature of an amorphous blend as described by the principle of Gordon-Taylor equation (Patterson et al., 2006; Yu, 2001; Fukuoka, Makita, and Yamamura, 1989; Craig et al., 1999). This indicated that  $E_2$  in solid dispersion examined by MTDSC with heating program II existed in an amorphous state.

In DSC curve of  $E_2$ ,  $T_g$  of  $E_2$  was observed during the second heating run. Thus, solid dispersion during the second heating run of heating program II analysis contained amorphous  $E_2$  blended with ERS. In comparison with the resulting  $T_g$  of blends obtained from MTDSC using heating program I,  $T_g$  behavior deviated from the nature of an amorphous blend. This indicated that  $E_2$  in solid dispersion during analysis with heating program I existed in a crystalline state. Thus, solid dispersion containing  $E_2$  at concentration higher than its solubility in ERS obtained from co-evaporation was composed of crystalline  $E_2$  blended with ERS, which was not described by the principle of Gordon-Taylor equation. This corresponded to the existing of melting point of crystalline  $E_2$  in solid dispersion examined by MTDSC with heating program I as shown in Figure 4.7. Pure crystalline  $E_2$  manifested its melting point at 179.88°C. The melting point of crystalline  $E_2$  decreased when the concentration of ERS in solid dispersion increased. This is the nature of crystalline component blended with an amorphous component

(a)



(b)

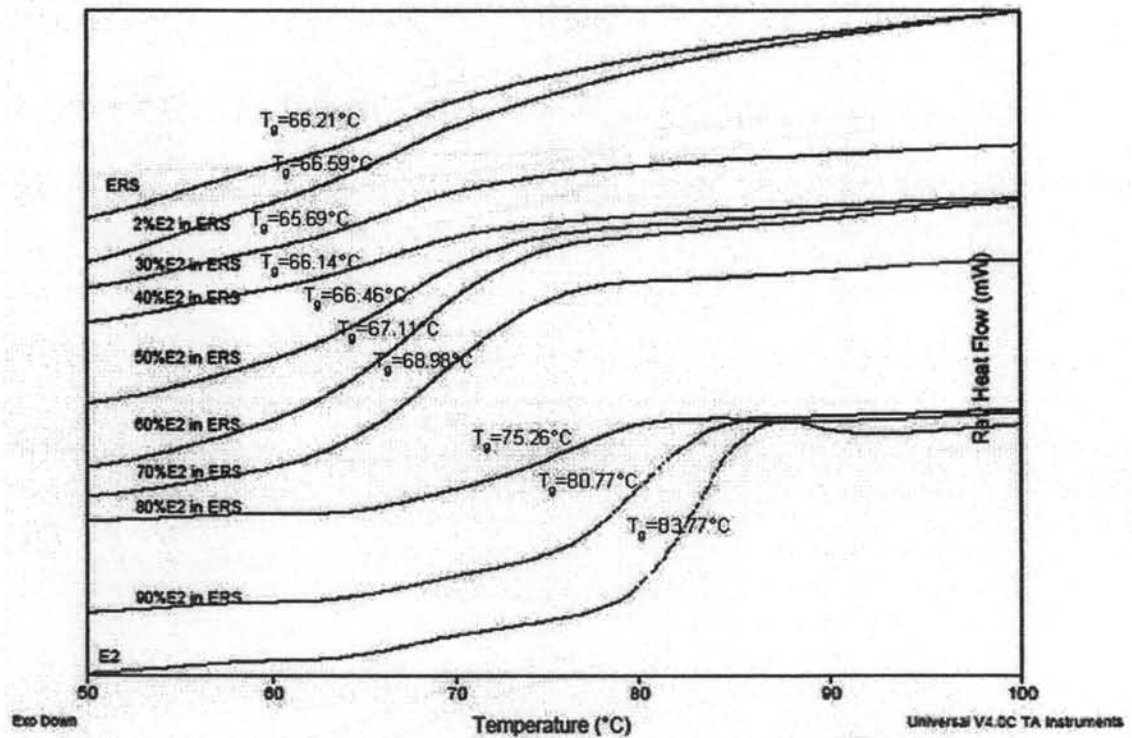
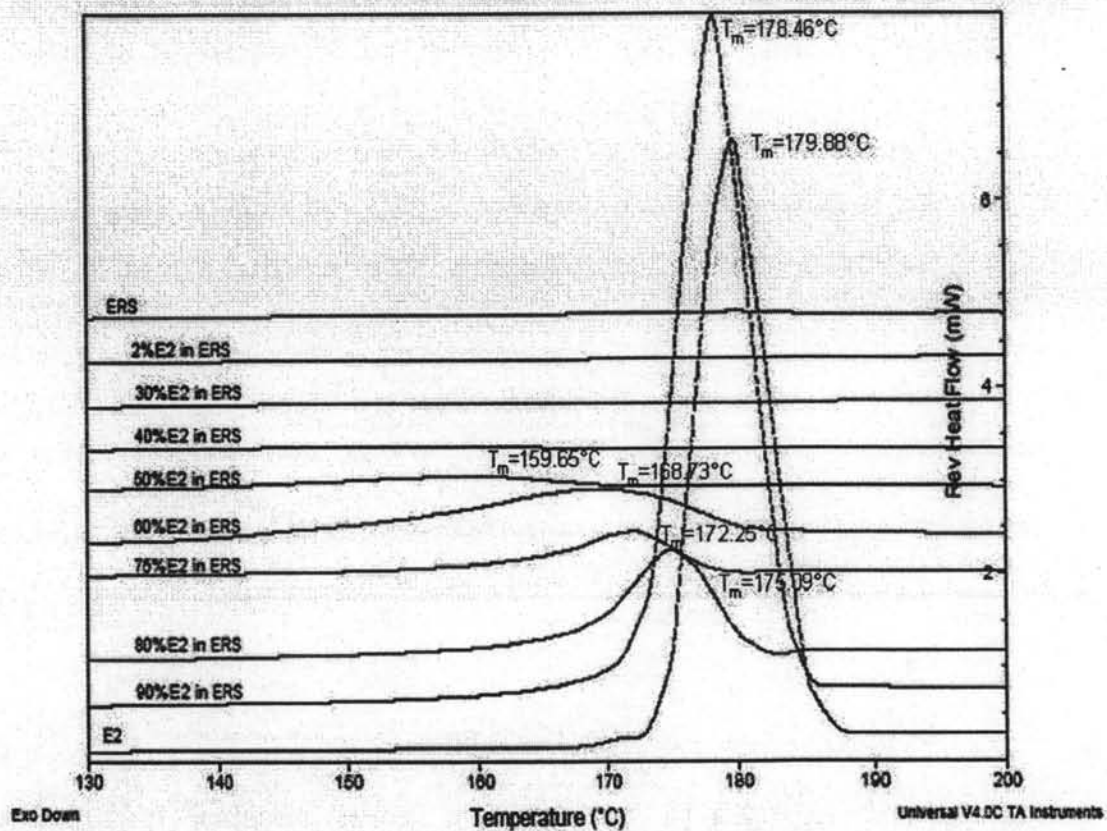


Figure 4.6  $T_g$  behavior of E<sub>2</sub> in ERS solid dispersions at concentration range of 0-100 % w/w obtained from MTDSC: (a) heating program I; (b) heating program II



**Figure 4.7** The melting point depression of E<sub>2</sub> in ERS solid dispersions at concentration range of 0-100 % w/w obtained from MTDSC scanning with the heating program I

(Tantishaiyakul et al., 1996; Kuo and Chang, 2001; Kuo et al., 2001; Satit Puttipatkhachorn et al., 2001). The addition of ERS as an amorphous component decreased the chemical potential of crystalline E<sub>2</sub> and led to a reduction of melting point explained by the Flory-Huggins theory (Rostami, 2000; Pimbert et al., 2002).

At lower weight percent of E<sub>2</sub> in solid dispersion, T<sub>g</sub> of a blend examined by MTDSC using either heating program I or heating program II were not different from T<sub>g</sub> of ERS. This phenomenon could also be explained by the principle of Gordon-Taylor equation. The Gordon-Taylor equation predicts T<sub>g</sub> of an amorphous blend based on weight fractions (w) and T<sub>g</sub> of its components and allows K to be calculated using amorphous density (ρ) as shown in previously described equation (2.16) and (2.17), respectively (Patterson et al., 2006; Craig et al., 1999; Schneider, 1988).

$$T_g = \frac{w_1 T_{g1} + w_2 T_{g2}}{w_1 + K w_2} \quad 2.16$$

$$K = \frac{\rho_1 T_{g1}}{\rho_2 T_{g2}} \quad 2.17$$

Thus, effect of amorphous E<sub>2</sub> properties on the resulting T<sub>g</sub> of a blend was trivial. T<sub>g</sub> of solid dispersions containing lower weight percent of E<sub>2</sub> were nearly similar to T<sub>g</sub> of ERS.

#### 4.4 Conclusions

E<sub>2</sub> existed in an amorphous state in solid dispersion containing 1 and 2 % w/w E<sub>2</sub> in ERS. E<sub>2</sub> dissolution rate did not act as the limiting resistance to drug release from implants containing these solid dispersions. These implants released E<sub>2</sub> in a way that deviated from the zero-order kinetics. In solid dispersion containing 10, 20 and 30 % w/w E<sub>2</sub> in ERS, E<sub>2</sub> existed in a crystalline state. E<sub>2</sub> dissolution rate offered as the limiting resistance to drug release from implants containing these solid dispersions. E<sub>2</sub> released from these implants could be described by the zero-order model.