CHAPTER IV RESULTS AND DISCUSSION

4.1 Effect of Catalysts on Hydrogen Desorption in Li-N-H Systems

4.1.1 Mixed by Mortar and Pestle

A mixture of LiNH₂ and LiH (1:1 mol ratio) was added with 1 mol% TiO₂ or Fe and mixed by using mortar and pestle for 15 min. Effects of the catalysts on hydrogen desorption are shown in Figure 4.1. The amounts of released hydrogen from all samples are relatively the same, about 0.4 – 0.5 wt% at 200°C. In addition, either catalyst barely affects the hydrogen desorption kinetics.

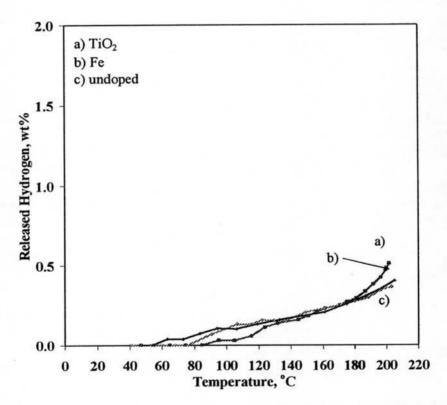


Figure 4.1 Correlation between temperature and hydrogen released during hydrogen desorption from the mixture of LiNH₂ and LiH with a 1:1 mol ratio by mortar and pestle for 15 min: a) 1 mol% TiO₂, b) 1 mol% Fe, and c) undoped.

4.1.2 Mixed by Agate Ball Milling

A 1:1 mol ratio of LiNH₂ and LiH mixture was added with 1 mol% TiO₂, ZrCl₄, Ni, or Fe and mixed by agate ball milling to compare the effect of the catalysts on hydrogen desorption, as shown in Figure 4.2. The results show that the catalysts increase the amount of hydrogen desorption and increase the hydrogen desorption kinetics. The amounts of released hydrogen from the doped samples are significantly higher than the undoped one, which is 0.43 wt%. The samples doped with ZrCl₄, Ni and Fe desorb 0.78, 0.76 and 0.68 wt% hydrogen at 200°C, respectively, and give the same rate of hydrogen desorption. However, the sample doped with TiO₂ desorbs 1.5 wt% hydrogen at the same temperature and clearly increases the hydrogen desorption kinetics.

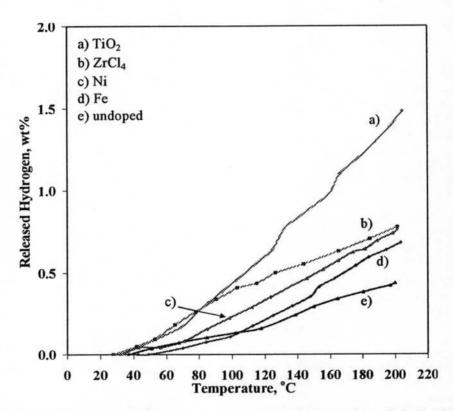


Figure 4.2 Correlation between temperature and hydrogen released during hydrogen desorption from the mixture of LiNH₂ and LiH with a 1:1 mol ratio by using agate ball milling for 2 h: a) 1 mol% TiO₂, b) 1 mol% ZrCl₄, c) 1 mol% Ni, d) 1 mol% Fe, and e) undoped.

4.2 Effect of Mixing Means on Hydrogen Desorption in Li-N-H Systems

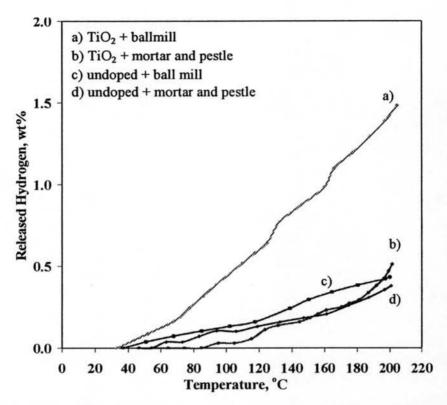


Figure 4.3 Correlation between temperature and hydrogen released during hydrogen desorption from the mixture of LiNH₂ and LiH with a 1:1 mol ratio: a) 1 mol% TiO₂ by using agate ball milling for 2 h, b) 1 mol% TiO₂ by using mortar and pestle for 15 min, c) undoped by using agate ball milling for 2 h, and d) undoped by using mortar and pestle for 15 min.

Figure 4.3 shows the temperature program desorption of the mixture of LiNH₂ and LiH with a 1:1 mol ratio with different mixing techniques. It was found that the sample doped with 1 mol% TiO₂ (Figure 4.3a) has an outstandingly high amount of released hydrogen, about 1.5 wt% while the amount of released hydrogen of the TiO₂ doping sample mixed by using mortar and pestle (Figure 4.3b) is about 0.5 wt%. In the case of the undoped samples (Figures 4.3c and 4.3d), the samples seem to release hydrogen at the same rate. However, the one mixed by ball milling desorbs hydrogen at a lower temperature because the ball milling can effectively reduce an average particle size and the small particle has a high surface area to release hydrogen. It is clear that not only a catalyst but also a mixing method affect

the hydrogen desorption of the mixture of LiNH₂ and LiH. The samples doped with Fe was also studied for its effect but the hydrogen desorption of the samples is not distinguishable by the means of mixing.

After desorption, the samples were re-absorbed under the hydrogen pressure of 500 psi at 180°C for 12 h. It was found that all tested samples are not able to re-absorb hydrogen.

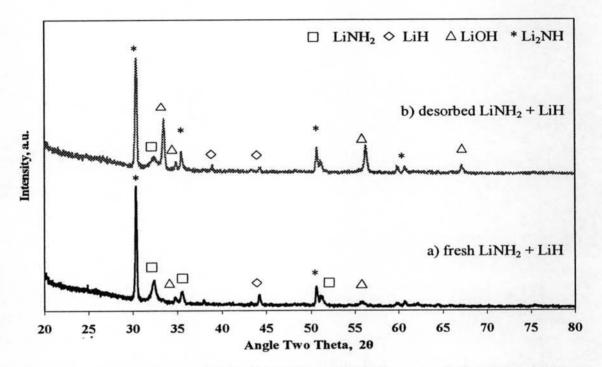


Figure 4.4 XRD patterns of a) fresh LiNH₂ and LiH after agate ball milling and b) desorbed LiNH₂ and LiH at 200°C.

To find out why the hydrogen desorption of the LiNH₂ and LiH mixture is not as high as that reported by Ichikawa et al. (2005) (up to 6.5 wt%), characterization by a Rigaku X-ray diffractrometer at room temperature over a range of diffraction angle from 20 – 80 with CuK-alpha radiation (40 kV, 30 mA) is used. Due to the air sensitivity of the samples, before the X-ray diffraction analysis, they were covered with a Kapton tape. Figure 4.4 shows XRD patterns of the mixture of LiNH₂ and LiH: a) fresh sample and b) desorbed sample. Results from the XRD patterns supports the results from the temperature program desorption. They show that the main peaks of the fresh sample are Li₂NH, which is a product from the

reaction between LiNH₂ and LiH: LiNH₂ + LiH ↔ Li₂NH + H₂ The other peaks with lower intensity indicate LiOH and LiNH₂. This implies that the sample has been decomposed before the desorption, possibly during the ball milling. In addition, the sample has many peaks of impurities that would consequently cause the low hydrogen desorption of the LiNH₂ and LiH mixture.

To examine the peaks from the XRD pattern of the fresh mixture of LiNH₂ and LiH (Figure 4.4a), as-received LiNH₂, fresh LiNH₂ after agate ball milling, and as-received LiH were characterized by XRD. Figure 4.5 shows XRD patterns of the undoped LiNH₂ sample: a) as-received sample; b) milled sample. The XRD patterns show that the main peaks of fresh LiNH₂ after ball milling (Figure 4.5b) are Li₂NH and intensity of LiNH₂ decreases compared with the as-received one (Figure 4.5a). This substantiates that LiNH₂ is decomposed to Li₂NH during the milling.

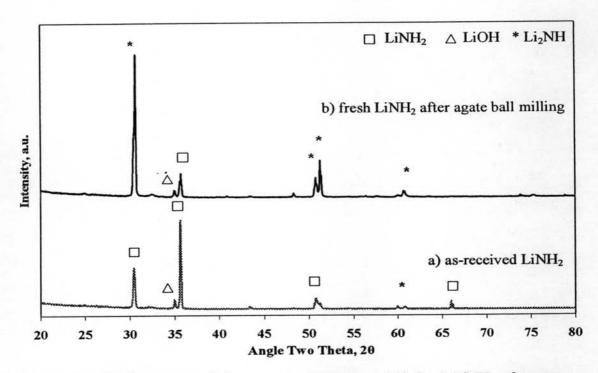


Figure 4.5 XRD patterns of a) as-received LiNH₂ and b) fresh LiNH₂ after agate ball milling for 2 h.

The XRD pattern of as-received LiH in Figure 4.6 shows very high intensity of LiOH, which occurs by reacting with air or moisture. LiH in this system is not pure enough for further study of the LiNH₂ and LiH reaction.

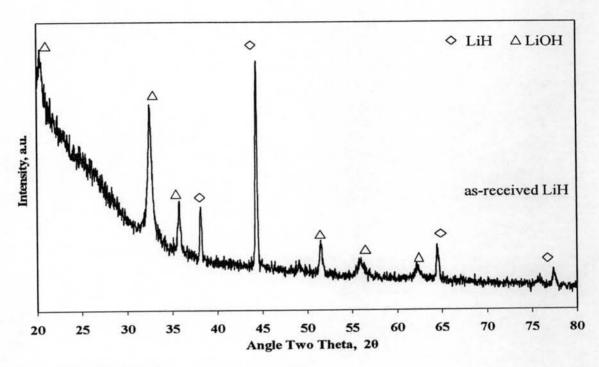


Figure 4.6 XRD patterns of as-received LiH.

4.3 Effect of LiAlH4 on Hydrogen Desorption in LiNH2

XRD patterns of the undoped LiAlH₄ sample at different conditions: a) asreceived, b) fresh sample after milling, and c) desorbed sample at 200°C, are
illustrated in Figure 4.7. After the milling, intensity of LiAlH₄ decreases and some
peaks of LiAlH₄ disappear. Progressive hydrogen desorption at 200°C shows
complete desorption of the hydride to LiH and Al. Furthermore, the increase in the
intensity and narrow peak of aluminum and LiH show the formation of their
crystallites. LiH in higher intensity can also be seen. Because of the decomposition
of LiAlH₄ to LiH, LiAlH₄ was used as a source of LiH to react with LiNH₂.

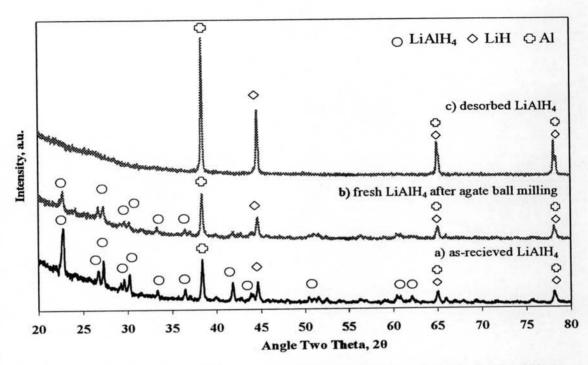


Figure 4.7 XRD patterns of a) as-received LiAlH₄, b) fresh LiAlH₄ after milling, and c) desorbed LiAlH₄ at 200°C.

Figure 4.8 shows the hydrogen desorption of 1:1, 1:3, and 1:5 mol ratios of LiNH2 and LiAlH4 compared with pure LiAlH4 and pure LiNH2. The result shows that pure LiAlH4 gives the highest hydrogen evolution rate when temperature reaches about 120°C and releases about 9.3 wt% hydrogen at 200°C followed by the 1:5 mol ratio sample, 5.0 wt%. The amounts of hydrogen desorption of samples with 1:3 and 1:1 mol ratios are almost the same, about 2.5 wt%. With the increase in the LiAlH4 loading, the hydrogen desorption behavior is similar to that of pure LiAlH4. However, the amount of hydrogen desorption of the 1:5 mol ratio is less than that of pure LiAlH4. In this case, the explanation is that LiNH2 in the mixture may somehow affect the decomposition of LiAlH4 during the milling. The XRD patterns in Figures 4.7b and 4.9a confirm that the fresh 1:5 mol ratio sample (Figure 4.9a) has lower peaks of LiAlH₄ than those of pure LiAlH₄ (Figure 4.7b) meaning that LiNH₂ in the 1:5 mol ratio sample has an effect on the decomposition of LiAlH₄. Comparison among the decomposition slopes, the three mixtures of LiNH2 and LiAlH4 with pure LiNH2, shows that the formers are not as steep as the latter. A possible reason may be, during the decomposition of LiNH2 to NH3, NH3 is consumed by the reaction with LiH from LiAlH₄ resulting in a lower total pressure. In the case of the 1:5 mol ratio sample, its slope at low temperatures is still lower than that of pure LiNH₂. However, as the temperature is high enough, excess LiH is formed from the decomposition of LiAlH₄ resulting in the higher reaction rate between LiH and NH₃. And because of that, its slope is not as high as that of LiAlH₄ decomposition.

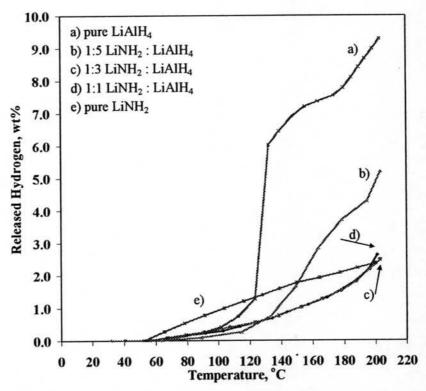


Figure 4.8 Temperature program desorption from room temperature to 200°C with the heating rate of 2°C min⁻¹ by using agate ball milling for 1 h: a) pure LiAlH₄ b) 1:5 mol ratio of LiNH₂ and LiAlH₄ c) 1:3 mol ratio of LiNH₂ and LiAlH₄ d) 1:1 mol ratio of LiNH₂ and LiAlH₄, and e) pure LiNH₂.

XRD patterns of fresh and desorbed 1:5 mol ratio of LiNH₂ and LiAlH₄ in Figure 4.9 show that the fresh sample (Figure 4.9a) still has some peaks of LiAlH₄ to decompose to LiH and then reacts with LiNH₂. However, based on the results, it is not possible to specify whether hydrogen is from the composition of LiAlH₄ or reaction of the LiNH₂ and LiH, or both. XRD patterns of fresh and desorbed samples with 1:1 and 1:3 mol ratios of LiNH₂ and LiAlH₄ are the same.

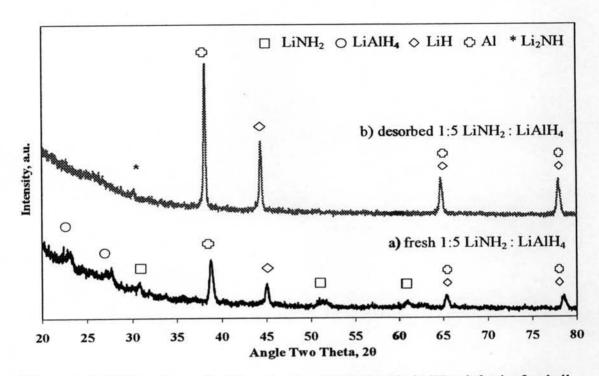


Figure 4.9 XRD patterns of a 1:5 mol ratio of LiNH₂ and LiAlH₄: a) fresh after ball milling and b) desorbed at 200°C.

4.4 Effect of Catalysts on Hydrogen Desorption in Li-Al-N-H Systems

4.4.1 Effect of Catalysts

To study the effect of catalyst types on hydrogen desorption from the mixture of LiNH₂ and LiAlH₄, ZrCl₄ and TiO₂ were used. Zidan *et al.* (1999) reported that Zr and Ti affected the hydrogenation/dehydrogenation of NaAlH₄ so some physical properties and mechanisms of Zr and Ti may affect LiAlH₄. Figure 4.10 shows the hydrogen desorption of the 1:5 mol ratio of LiNH₂ and LiAlH₄ doped with 4 mol% ZrCl₄ or TiO₂ compared with the undoped one mixed with agate ball milling. The figure clearly confirms that the catalysts enhance the kinetics of the hydrogen desorption. The ZrCl₄ doped sample (Figure 4.10a) shows a higher amount of hydrogen desorbed, 8.0 wt% at 200°C, than the sample doped with TiO₂ (Figure 4.10c) and the undoped one (Figure 4.10b), both around 5.0 wt%. The addition of ZrCl₄ increases the amount of hydrogen desorption and desorption kinetics while TiO₂ only increases the desorption kinetics.

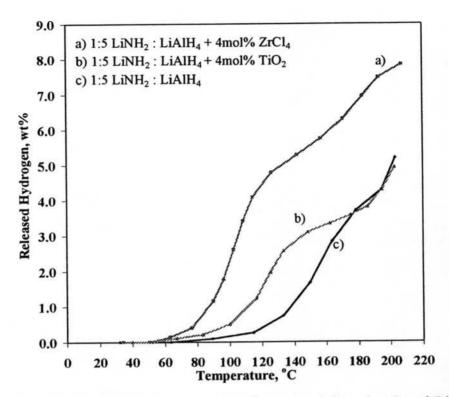


Figure 4.10 Temperature program desorption of a 1:5 mol ratio of LiNH₂ and LiAlH₄ mixed by using agate ball milling for 1 h: a) doped with 4 mol% ZrCl₄, b) doped with 4 mol% TiO₂, and c) undoped.

XRD results of 1:5 LiNH₂:LiAlH₄ doped with 4 mol% TiO₂ or 4 mol% ZrCl₄ after hydrogen desorption are shown in Figures 4.11 and 4.12, respectively. Comparison of the XRD patterns of the undoped sample (Figure 4.9) and the one doped with TiO₂ or ZrCl₄ shows that only the sample doped with ZrCl₄ resulted in the presence of LiCl after the milling. This may be due to the reaction of LiH with ZrCl₄ when the amount of ZrCl₄ is high enough, 4 mol% for this case, to form LiCl. After complete desorption, the XRD results show the increase in the formation of LiH and Al for all samples (Figures 4.9b, 4.11b, and 4.12b). Lastly, we do not observe any peaks of the transition metal compound (Zr and Ti) in the samples after the hydrogen desorption/absorption by the X-ray diffraction technique.

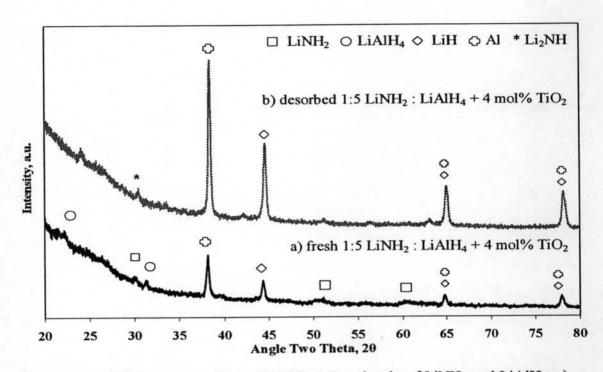


Figure 4.11 XRD patterns of 4 mol% TiO₂-1:5 mol ratio of LiNH₂ and LiAlH₄: a) fresh after ball milling and b) desorbed at 200°C.

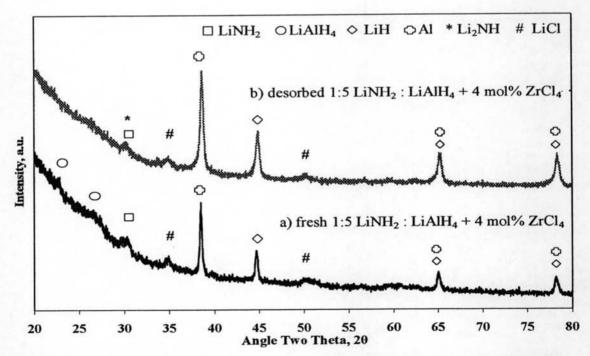


Figure 4.12 XRD patterns of 4 mol% ZrCl₄-1:5 mol ratio of LiNH₂ and LiAlH₄: a) fresh after ball milling, b) desorbed at 200°C, and c) absorbed at 180°C.

4.4.2 Effect of ZrCl₄ Loading

Amounts of ZrCl₄ doped on the 1:5 mol ratio of LiNH₂ and LiAlH₄ are at 1 and 4 mol%. Its dehydrogenation from room temperature to 200°C with the heating rate of 2°C min⁻¹ is shown in Figure 4.13. From the figure, only a small amount of ZrCl₄ can enhance the desorption kinetics. The higher amount of ZrCl₄ loading increases the desorption kinetics and hydrogen capacity. However, although the desorption temperature of the samples with the two loading are relatively the same, they are lower than that of the undoped sample. At the 4 mol% ZrCl₄ loading, the hydride starts to decompose at 55°C and releases 8.0 wt% hydrogen at 200°C.

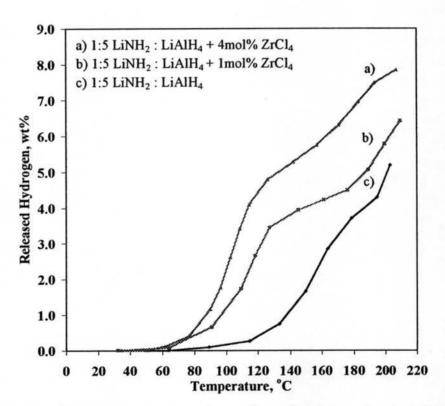


Figure 4.13 Temperature program desorption of a 1:5 mol ratio of LiNH₂ and LiAlH₄ mixed by using agate ball milling for 1 h: a) doped with 4 mol% ZrCl₄, b) doped with 1 mol% ZrCl₄, and c) undoped.

4.4.3 Reversibility of Li-Al-N-H Systems

stability of hydrogen reversibility and To study the desorption/absorption of the mixture of LiAlH4 and LiNH2, a sample is dehydrogenated from room temperature to 200°C with the heating rate of 2°C min⁻¹. The first dehydrogenation of 1:1 LiNH2 and LiAlH4 releases about 2.6 wt% hydrogen. After the first desorption, the sample was reabsorbed under 1,500 psi of hydrogen pressure at 180°C for 12 h. The amount of released hydrogen drops to 0.4 wt% for the subsequent desorptions, as shown in Figures 4.14. We observed that the decomposition temperature has been shifted from 50°C to 70°C after the first cycle. For other experiments of LiNH2 and LiAlH4 mixtures such as the 1:1 mol ratio doped with 1 mol% ZrCl₄, or 1:3, or 1:5 undoped or doped with 1 and 4 mol% ZrCl₄, only 0.15 - 0.4 wt% can be reabsorbed.

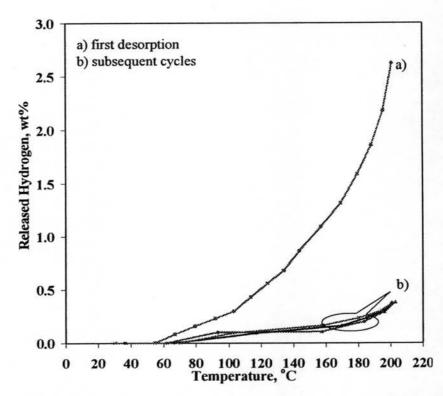


Figure 4.14 Temperature program desorption from the mixture of LiNH₂ and LiAlH₄ with a 1:1 mol ratio by using agate ball milling for 1 h: a) first cycle and b) subsequent cycles.

The total reversibility of the desorbed samples could not be achieved to their original states. The assumption for the reversibility of the Li-Al-N-H system is from the reaction between LiNH₂ and LiH but after absorption (Figure 4.15c), the LiNH₂ disappears and only Al and LiH are present. XRD patterns of other studied samples with different LiNH₂ to LiAlH₄ ratios and different catalysts are more or less the same.

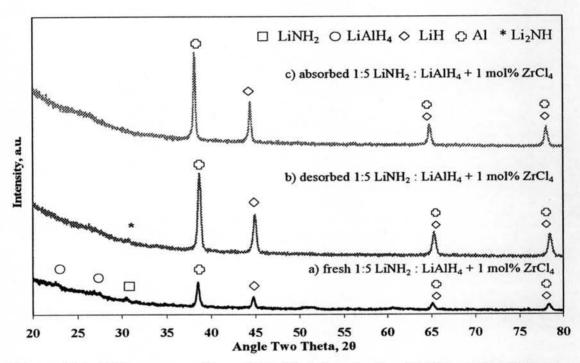


Figure 4.15 XRD patterns of 1 mol% ZrCl₄-1:5 mol ratio of LiNH₂ and LiAlH₄: a) fresh after ball milling, b) desorbed at 200°C, and c) absorbed at 180°C.

4.4.4 Effect of Sample Loading Methods

As NH₃ for the decomposition of LiNH₂ is expected to react with LiH from LiAlH₄, different sample loading methods were studied. The method is similar to Ichikawa *et al.* (2004). Here, three configurations were used. Configuration A is to place LiAlH₄ milled with 4 mol% ZrCl₄ on top of LiNH₂ milled with 4 mol% ZrCl₄ with a 1:5 mol ratio of LiNH₂ to LiAlH₄. Configuration B is to place LiAlH₄ milled with 4 mol% ZrCl₄ on top of LiNH₂ with a 1:5 mol ratio of LiNH₂ to LiAlH₄. In configuration C, 1:5 LiNH₂:LiAlH₄ doped with 4 mol% ZrCl₄ is used without the separation of each hydride. Figure 4.16 shows the configurations of a 1:5 mol ratio of

LiNH₂ and LiAlH₄. We can see that the first two configurations are based on the postulation that NH₃ from LiNH₂ passes through the bed of LiAlH₄ and react with LiH to form H₂ while the last configuration is to mix these constituents homogeneously.

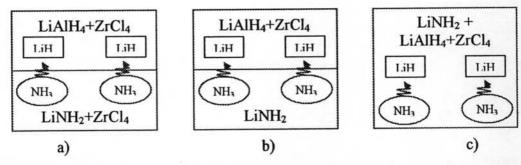


Figure 4.16 Packing configuration of LiNH₂ and LiAlH₄ mixtures: a) two-layered sample both of which doped with 4 mol% ZrCl₄, b) two-layered sample and only LiAlH₄ doped with 4 mol% ZrCl₄, and c) both hydrides mixed with 4 mol% ZrCl₄.

Figure 4.17 shows the temperature program desorption between different sample loading methods. For the configurations A and B, it should be noted that the shape of the hydrogen emission curve in Figure 4.17a is quite similar to that of Figure 4.17b. In addition, the amounts of released hydrogen are not significantly different. For the LiNH₂ phase, the doping of ZrCl₄ barely changes the hydrogen desorption behavior. In other words, ZrCl₄ does not have any influence on the hydrogen desorption or desorption kinetics on LiNH₂. By comparing the results of the two-layered samples (Figures 4.17a and 4.17b) and one-layered one (Figure 4.17c), it indicates that the contact time is one important parameter to improve the hydrogen desorption. NH₃ from LiNH₂ should have long enough resident time to react with LiH from LiAlH₄. In the case of LiNH₂ mixed together with LiAlH₄ (Figure 4.17c), LiNH₂ in the mixture has an effect on the decomposition of LiAlH₄ during the milling, as explained in Figure 4.8, Ichikawa *et al.* (2004b) concluded that LiH plays a role for the decomposition of LiNH₂ from the study between LiNH₂ and LiH compared with two-layer undoped LiNH₂ and LiH.

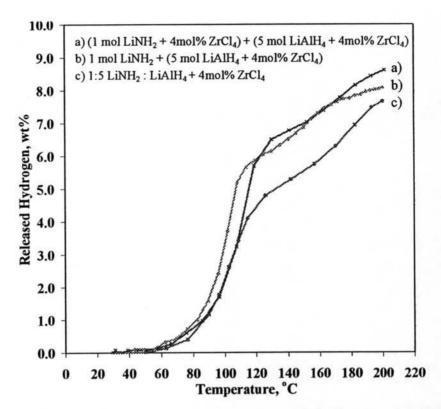


Figure 4.17 Correlation between temperature and hydrogen capacity, during hydrogen desorption on a 1:5 mol ratio of LiNH₂ and LiAlH₄: a) two-layered sample each of which doped with 4 mol% ZrCl₄, b) two-layered sample only LiAlH₄ doped with 4 mol% ZrCl₄, and c) one-layered sample doped with 4 mol% ZrCl₄.

Effects of different catalysts on the two-layered 1:1 LiNH₂ and LiAlH₄ doped with 4 mol% VCl₃ are shown in Figure 4.18a, and the two-layered doped with 4 mol% ZrCl₄ (Figure 4.18b) shows higher amounts of hydrogen desorption and reaction rates than those of the one-layered (Figure 4.18c). It is not only caused by the effect of the transition metals but also caused by the packing configuration. VCl₃ seems to be the most effective additive to increase the hydrogen desorption and the desorption kinetics.

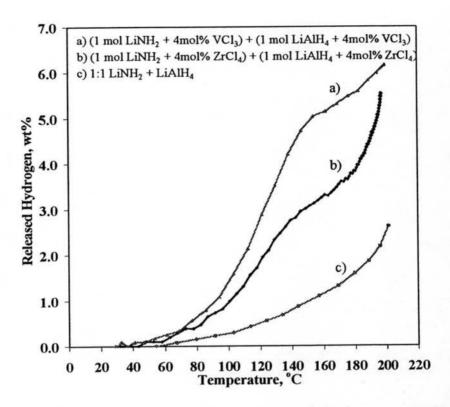


Figure 4.18 Correlation between temperature and hydrogen capacity, during hydrogen desorption on a 1:1 mol ratio of LiNH₂ and liAlH₄: a) two-layered sample which are doped with 4 mol% VCl₃, b) two-layered sample which are doped with 4 mol% ZrCl₄, and c) undoped.

XRD patterns of each layer, LiNH₂ doped with 4 mol% VCl₃ and LiAlH₄ doped with 4 mol% VCl₃, are shown in Figures 4.19 and 4.20, respectively. The samples of these figures are doped with 4 mol% of chloride forms, like Figure 4.12, then LiCl are observed and any peaks of the transition metal compounds, V or Zr, in the samples are not seen. The XRD patterns explain the results from Figure 4.17 that the transition metals have an effect on the hydrogen desorption. While the XRD patterns of the hydrogen absorption of all samples are not different as shown in Figures 4.15, hydrogen can be desorbed in the subsequent cycle in the range of 0.15 to 0.4 wt%.

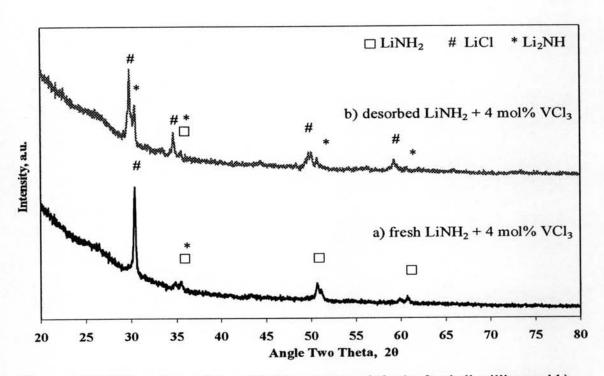


Figure 4.19 XRD patterns of 4 mol% ZrCl₄-LiNH₂: a) fresh after ball milling and b) desorbed at 200°C.

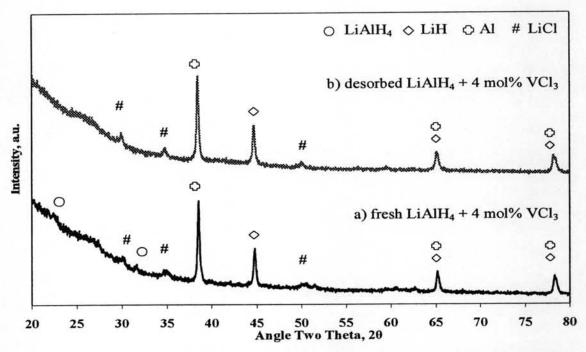


Figure 4.20 XRD patterns of 4 mol% ZrCl₄-LiAlH₄: a) fresh after ball milling and b) desorbed at 200°C.

4.5 TPD and MS of LiNH2 and LiNH2/LiAlH4

4.5.1 Effect of Catalyst on LiNH₂

Figures 4.21 shows the TPD-MS patterns of LiNH₂ milled with and without 4 mol% ZrCl₄ for 1 h. The undoped sample desorbs ammonia (m/z = 17) between 250 - 700°C with a significant peak at 500°C. For LiNH₂ doped with 4 mol% ZrCl₄, ammonia desorbs in the temperature range of 150 - 650°C and the intensity of ammonia is much lower than that of the undoped sample.

Figure 4.22 shows the hydrogen desorption from LiNH₂ milled with and without 4 mol% ZrCl₄. The undoped sample desorbs hydrogen (m/z = 2) between 450 - 650°C while the sample doped with 4 mol% ZrCl₄ desorbs hydrogen between 350 - 650°C and the relative intensity is higher than that of the undoped sample.

Results from both figures show that the catalyst decreases the temperatures of the ammonia emission while it does not significantly affect on the temperature of the hydrogen emission. The catalyst does not only decrease the ammonia emission, it also increases the hydrogen desorption from LiNH₂.

4.5.2 Effect of LiAlH4 Loading

Comparison of the TPD-MS between the ball milled LiNH₂ and a 1:1 mol ratio of LiNH₂ and LiAlH₄ (Figure 4.23) shows that the 1:1 mol ratio sample decomposes at significantly lower temperatures than the ball milled LiNH₂. The mixture starts to desorb ammonia at 100 °C and completes the desorption around 550°C. This may be due to LiH from LiAlH₄ accelerates the decomposition of LiNH₂. This result is consistent with that of Ichkawa *et al.* (2004b).

Decomposition of 1:1 and 1:5 mol ratios of LiNH₂ and LiAlH₄ is shown in Figures 4.24 to 4.26. Figure 4.24 shows that the 1:5 mol ratio sample desorbs ammonia in the temperature range of 100 – 550°C with high ammonia intensity at 200, 300 and 550°C while the 1:1 mol ratio one desorbs ammonia with much less intensity.

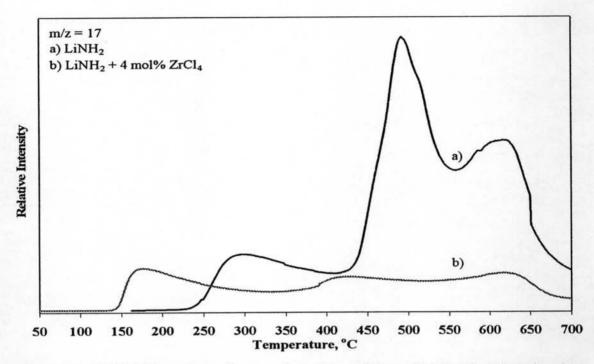


Figure 4.21 TPD-MS patterns of ammonia (m/z = 17) from LiNH₂: a) undoped and b) doped with 4 mol% ZrCl₄.

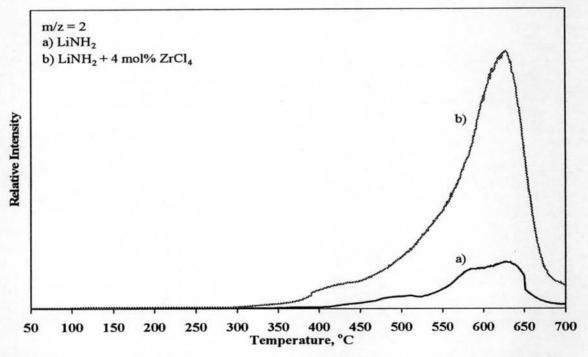


Figure 4.22 TPD-MS patterns of hydrogen (m/z = 2) from LiNH₂: a) undoped and b) doped with 4 mol% ZrCl₄.

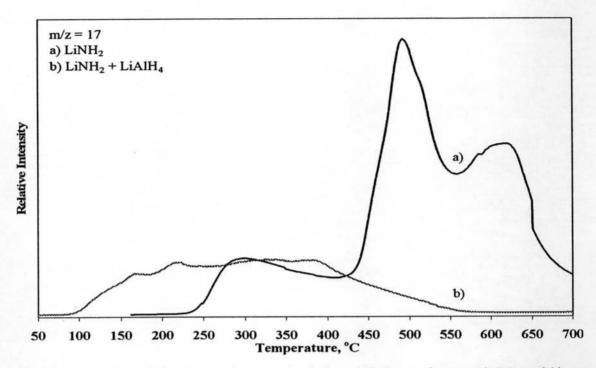


Figure 4.23 TPD-MS patterns of ammonia (m/z = 17) from: a) pure LiNH₂ and b) 1:1 mol ratio of LiNH₂ and LiAlH₄.

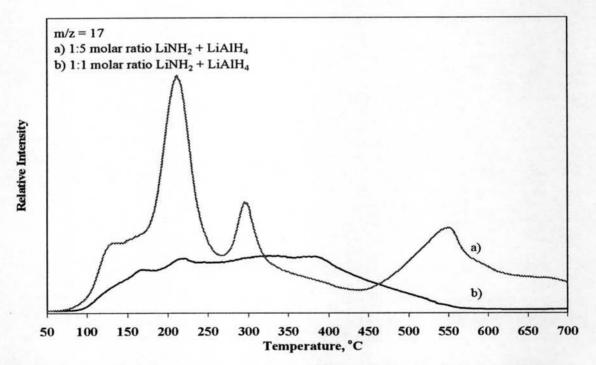


Figure 4.24 TPD-MS patterns of ammonia (m/z = 17) from the mixture of LiNH₂ and LiAlH₄: a) 1:5 mol ratio and b) 1:1 mol ratio.

Figure 4.25 shows the TPD-MS patterns of a 1:1 mol ratio of LiNH₂ and LiAlH₄. The main peak of this pattern is ammonia, which desorbs between 100 – 550°C while the hydrogen intensity is much low. The TPD-MS patterns of a 1:5 mol ratio of LiNH₂ and LiAlH₄ in Figure 4.26 show that the increase in the LiAlH₄ loading decreases the ammonia emission and increases the hydrogen intensity. It can be suggested that the higher hydrogen intensity may be from the decomposition of LiAlH₄ and the reaction between NH₃ and LiH, which is evidenced by the decrease in the ammonia intensity.

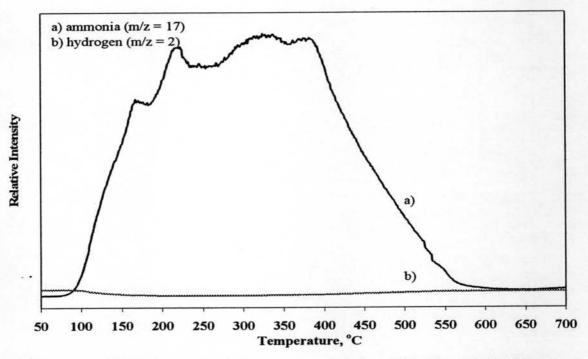


Figure 4.25 TPD-MS patterns from a 1:1 mol ratio of LiNH₂ and LiAlH₄: a) ammonia and b) hydrogen.

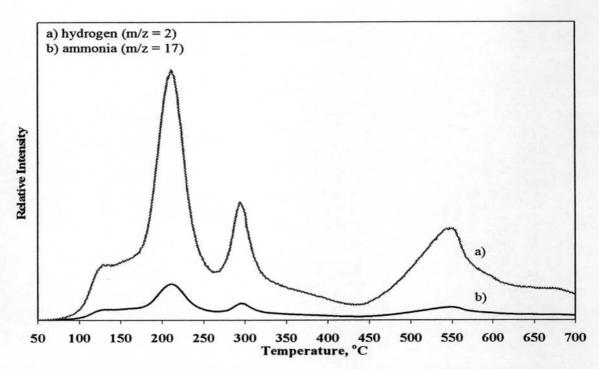


Figure 4.26 TPD-MS patterns from a 1:5 mol ratio of LiNH₂ and LiAlH₄: a) hydrogen and b) ammonia.

4.5.3 Effect of Catalyst on LiNH2/LiAlH4

Comparison between a 1:5 mol ratio of LiNH₂ and LiAlH₄ with and without 4 mol% ZrCl₄ is shown in Figure 4.27. It is clearly seen that ZrCl₄ decreases ammonia emission. The ammonia and hydrogen intensity of the 1:5 mol ratio of LiNH₂ and LiAlH₄ doped with 4 mol% ZrCl₄ is shown in Figure 4.28. It shows the lower desorption temperature for both ammonia and hydrogen than that of the undoped sample (Figure 4.26). In this case, the catalyst decreases the desorption temperature of the sample.

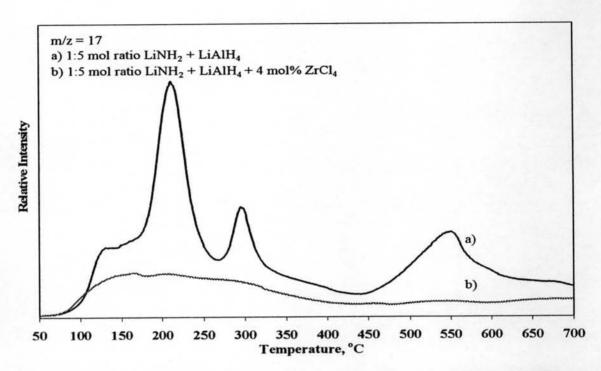


Figure 4.27 TPD-MS patterns of ammonia (m/z = 17) from a 1:5 mol ratio of LiNH₂ and LiAlH₄ by using agate ball milling for 1 h: a) undoped and b) doped with 4 mol% ZrCl₄.

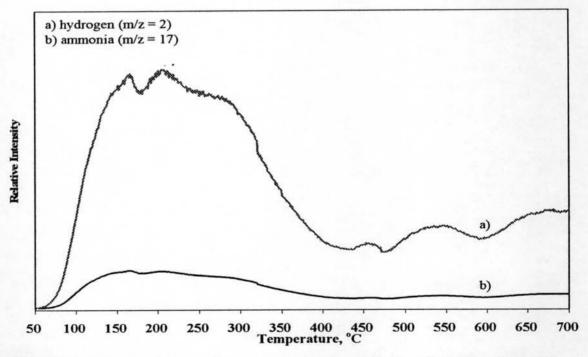


Figure 4.28 TPD-MS patterns from a 1:5 mol ratio of LiNH₂ and LiAlH₄ doped with 4 mol% ZrCl₄ by using agate ball milling for 1 h: a) hydrogen and b) ammonia.

4.5.4 Effect of Two-Layered Packing Sample

To confirm the reaction between ammonia from the decomposition of LiNH₂ and LiH from the decomposition of LiAlH₄ should have long enough contact time to result in H₂, the two-layered packing sample was studied. Figure 4.29 shows TPD-MS patterns of the two-layered on a 1:5 mol ratio of LiNH₂ and LiAlH₄, each of which is doped with 4 mol% ZrCl₄. The catalyst can decrease the desorption temperature, the highest peak of hydrogen is at 100°C and the two-layered packing can increase the hydrogen desorption.

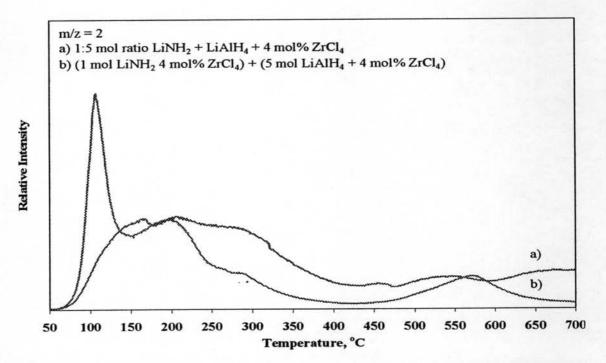


Figure 4.29 TPD-MS patterns of hydrogen (m/z = 2) from a 1:5 mol ratio of LiNH₂ and LiAlH₄ doped with 4 mol% $ZrCl_4$: a) two-layered sample each of which doped with 4 mol% $ZrCl_4$ and b) one-layered sample.